

## Molecular orbital calculation on the chemical bonding character of phosphate glasses

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### Abstract

Chemical bonding character of various bonds in phosphate glasses was investigated on the basis of bond overlap population which was evaluated from molecular orbital calculations. Cluster models were constructed from various phosphate crystals. There are two terminal P–O bonds in phosphates, that is, P=O double bonds and P–NBO (non-bridging oxygen) bonds. In bond overlap population, P=O bonds were much larger than P–NBO bonds. In P=O bonds, no significant anti-bonding overlap between P and O was observed at around the HOMO levels, which was also the case in Si–NBO bonds. In resonance state between P=O and P–NBO bonds, which was seen in  $(\text{PO}_{2/2}\text{O}_2)^- \text{Q}^2$  unit, anti-bonding overlaps were clearly confirmed, which reduced the bond overlap population of terminal P–O bonds. In P–O–M bridges (M = B, Si, Ge and Al), more electrons were populated on P–O bonds than M–O bonds, which was due to the larger electronegativity of P atoms.

### Introduction

Phosphate glasses have excellent physical properties such as low glass transition temperatures and high thermal expansion coefficients, and hence phosphate glasses have been applied to low-temperature sealing glasses. In pure  $\text{P}_2\text{O}_5$  glass, glass networks consist of tetrahedral  $\text{PO}_4$  units, where three oxygens are bridging oxygens (BOs) in P–O–P bridging bonds and the residual oxygen is doubly-bonded oxygen (DBO). The DBOs are located at the terminal of glass networks, which is similar to non-bridging oxygens (NBOs). In alkali or alkaline-earth oxides containing binary glasses, NBOs are also formed in the glasses. The NBOs are singly-bonded to phosphorous atoms, and hence they possess a negative unit charge. To compensate the negative charge of NBOs, alkali or alkaline-earth ions are bonded to NBOs to retain electroneutrality.

The authors' research group has studied electronic states and chemical bonding characters of various glasses mainly based on O1s X-ray photoelectron spectra (XPS) [1], where alkali phosphate glasses was also examined by XPS [2]. In the O1s XPS spectra, O1s signal was successfully separated into two components attributed to BO and DBO. After the alkali addition, however, a third new O1s component assigned to NBO was not observed. With increasing alkali content, the high binding energy component decreased and low binding energy component increased in relative intensity. The relative intensity of the low binding energy component was in good agreement with the total amount of DBO and NBO expected from the composition. It was consequently concluded that the DBO and NBO are in resonance state with each other [2]. Based on molecular orbital calculations, Uchino and Ogata reported that Li and Na ions

interacted equally with two terminal oxygens in a PO<sub>4</sub> unit but K and Rb ions interacted favorably with one of the two terminal oxygens [3], suggesting the resonance between DBO and NBO in lithium and sodium phosphate glasses and non-resonance in potassium and rubidium phosphate glasses. In either case, the terminal oxygens (TOs) are present in alkali or alkaline-earth phosphate glasses. TOs are very reactive as compared with BOs, and hence TO-abundant alkali phosphate glasses are easily soluble in water. Al<sub>2</sub>O<sub>3</sub> are often added to improve chemical durability of phosphate glasses; tetrahedral AlO<sub>4</sub> units are formed and enter into the glass network, consuming TOs. Further addition of Al<sub>2</sub>O<sub>3</sub> may be conducive to the formation of PO<sub>4</sub> units with no TOs, where PO<sub>4</sub> units with no TOs possess a unit positive charge, which is compensated by negatively-charged AlO<sub>4</sub> units. It is also the case in B<sub>2</sub>O<sub>3</sub> addition to phosphate glasses. The situation is somewhat different in SiO<sub>2</sub> addition; due to the small addition of SiO<sub>2</sub>, Si atoms enter into octahedral sites rather than tetrahedral sites.

Thus, in phosphate glasses, addition of various oxides leads to the structural changes not only of phosphate units but also of additives. At the same time, chemical bonding character of the glass constituents must be affected. In the present study, therefore, chemical bonding character of various phosphate systems was evaluated on the basis of molecular orbital calculations.

### Computational method and cluster models

Molecular orbital calculations were performed by using the discrete variational X $\alpha$  (DV-X $\alpha$ ) method [4], and chemical bonding characters were estimated by the Mulliken population analysis [5]. According to the Mulliken population analysis, chemical bonding character of a bond A–B was evaluated by the bond overlap population,  $Q_{AB}$ ,

$$Q_{AB} = 2 \sum_{i \in A, j \in B} Q_{ij}, \quad Q_{ij} = \sum_l Q_{ij}^l \quad (1)$$

$$Q_{ij}^l = n_l c_{il} c_{jl} S_{ij} \quad (2)$$

where  $Q_{ij}^l$  is the partial overlap population, that is, electronic population of the overlapped region between the atomic orbitals  $\chi_i$  and  $\chi_j$  in the molecular orbital  $\phi_l$ , which is the product of  $n_l$ : the number of electrons in MO  $\phi_l$  (usually  $n_l = 2$ ),  $c_{il}$  and  $c_{jl}$ : the contribution of atomic orbitals  $\chi_i$  and  $\chi_j$  in MO  $\phi_l$ , and  $S_{ij}$ : the overlap integral between  $\chi_i$  and  $\chi_j$ .

Cluster models were constructed from the various phosphate crystals; pure P<sub>2</sub>O<sub>5</sub> [6], alkali phosphates of Li<sub>3</sub>PO<sub>4</sub> [7], NaPO<sub>3</sub> [8], Na<sub>5</sub>P<sub>3</sub>O<sub>10</sub> [9], KPO<sub>3</sub> [10], and RbPO<sub>3</sub> [11], aluminophosphates of Al(PO<sub>3</sub>)<sub>3</sub> (Al = 6) [12], NaAlP<sub>2</sub>O<sub>7</sub> (Al = 6) [13], and AlPO<sub>4</sub> (Al = 4) [14], and phosphates containing network forming oxides of BPO<sub>4</sub> (B = 4) [15], SiP<sub>2</sub>O<sub>7</sub> (Si = 6) [16], and GeP<sub>2</sub>O<sub>7</sub> (Ge = 6) [17], where the numbers in the parentheses indicate the coordination number. As shown in Figure 1, oxygen atoms in the respective crystals were placed at the center of the clusters, and MO<sub>*n*</sub> polyhedra including the central oxygen were arranged around the central oxygen. The clusters were embedded in the Madelung potentials formed in the respective crystals to reduce the bond termination effect. The population analyses were done only for the bonds containing the central oxygens.

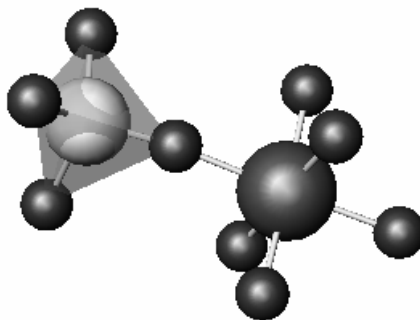


Figure 1: An example of cluster models.  
P(4)–O–Al(6) constructed from Al(PO<sub>3</sub>)<sub>3</sub> crystal.

## Results and Discussion

### Chemical Bonding Character of P<sub>2</sub>O<sub>5</sub> and Alkali Phosphates

In P<sub>2</sub>O<sub>5</sub> crystal [6], there are 2 phosphorous and 4 oxygen sites, in which the oxygen sites are classified into 2 BO and 2 DBO sites. Therefore, there are 4 distinct P–BO bonds and 2 distinct P=DBO bonds. As shown in Figure 2, the bond overlap population,  $Q_{PO}$  of P–BO bonds is widely distributed between 0.6 and 0.9, and it is also the case for P=DBO bonds. In a rough approximation, bond overlap population is inversely proportional to bond length. The wide distribution in  $Q_{PO}$  results from the wide distribution in P–O bond length;  $r(\text{P–BO}) = 0.1637, 0.1588, 0.1571, 0.1527$  nm and  $r(\text{P=DBO}) = 0.1492, 0.1426$  nm. As clearly seen in Fig. 2,  $Q_{PO}$  values of P=DBO bonds are larger than those of P–BO bonds, indicating that the P=O double bonds are stronger and more covalent than the P–O single bonds. The average  $Q_{PO}$  of P–BO bonds is 0.77, which is the largest among the conventional network formers of B<sub>2</sub>O<sub>3</sub> ( $Q_{B(3)O} = 0.64$ ), SiO<sub>2</sub> ( $Q_{Si(4)O} = 0.63$ ) and GeO<sub>2</sub> ( $Q_{Ge(4)O} = 0.54$ ) [18]. As for NBOs in sodium borate and silicate,  $Q_{B(3)O} = 0.84$  and  $Q_{Si(4)O} = 0.85$  [18];  $Q_{PO}$  of P–TO bonds in sodium phosphates are much larger than those of terminal bonds in borate and silicate, which is due to the strong P=O double bond.

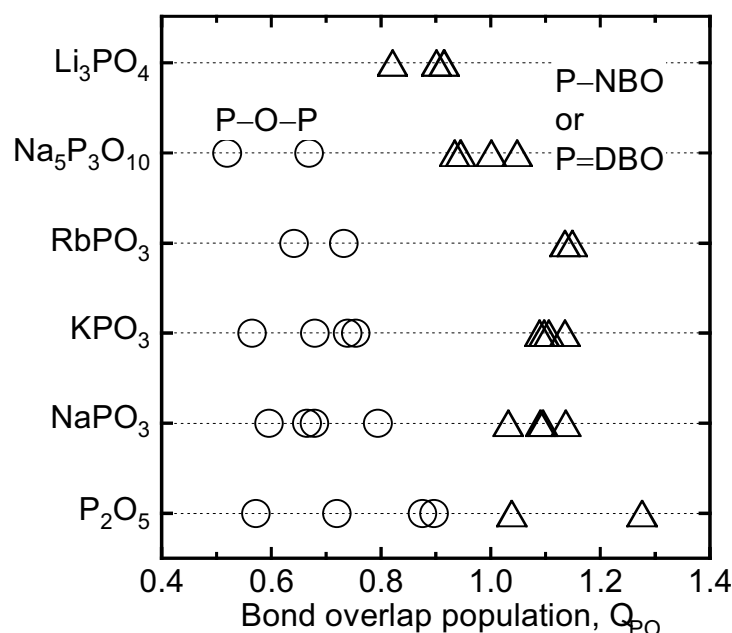


Figure 2: Bond overlap population,  $Q_{PO}$  of P–O bonds in

pure  $P_2O_5$  and alkali phosphate crystals.

In  $NaPO_3$  crystal [8], 4 of 6 oxygen sites are occupied by TOs, where the P–TO bond length narrowly ranges from 0.1484 to 0.1466 nm and TOs interact with 2 or 3 Na atoms at a distance between 0.2490 and 0.2350 nm. It is also the case in other alkali phosphate crystals. Thus, it is impossible to identify DBOs and NBOs in the alkali phosphate crystals examined in the present study. The bond orders of single and double bonds are 1 and 2, respectively. If P–NBO and P=DBO bonds were in non-resonant state, they would indicate different  $Q_{PO}$ . As shown in Fig. 2, however,  $Q_{PO}$  of P–TO bonds in  $NaPO_3$  is narrowly distributed, suggesting the resonance between P–NBO and P=DBO bonds. As shown in Fig. 2, it is also the case in  $KPO_3$  and  $RbPO_3$ ; in these crystals, terminal oxygen atoms are crystallographically equivalent, and hence NBO and DBO are indistinguishable.

With increasing the relative amount of alkali oxides, P–BO and P–TO bonds both decrease in  $Q_{PO}$ , which is probably associated with the increase in the relative amount of TOs. In the case of sodium silicate [19],  $Q_{SiO}$  of Si–BO bonds decreased and that of Si–NBO bonds increased with increasing alkali content. Na ions in silicates interact with not only NBOs but also BOs, and the contribution of anti-bonding overlap between the atomic orbitals of Si and BO increased with increasing the number of Na ions associated to BO. In the case of Si–NBO bonds, however, the anti-bonding overlap was not confirmed at around HOMO levels, and  $Q_{SiO}$  of Si–NBO bonds increased with the increase of alkali content, which was opposite to the change of  $Q_{PO}$  of P–TO bonds. As shown in Figure 3(a), only small contribution of anti-bonding overlap was confirmed in the P=O bond. As shown in Figs. 3(a) ~ (c), however, the anti-bonding overlaps between the atomic orbitals of P and O are clearly recognized both in P–BO and P–TO bonds at around HOMO levels (0 ~ –5 eV). The anti-bonding overlap in P–TO bonds in alkali phosphates may be caused by the resonance between P–NBO and P=DBO bonds.

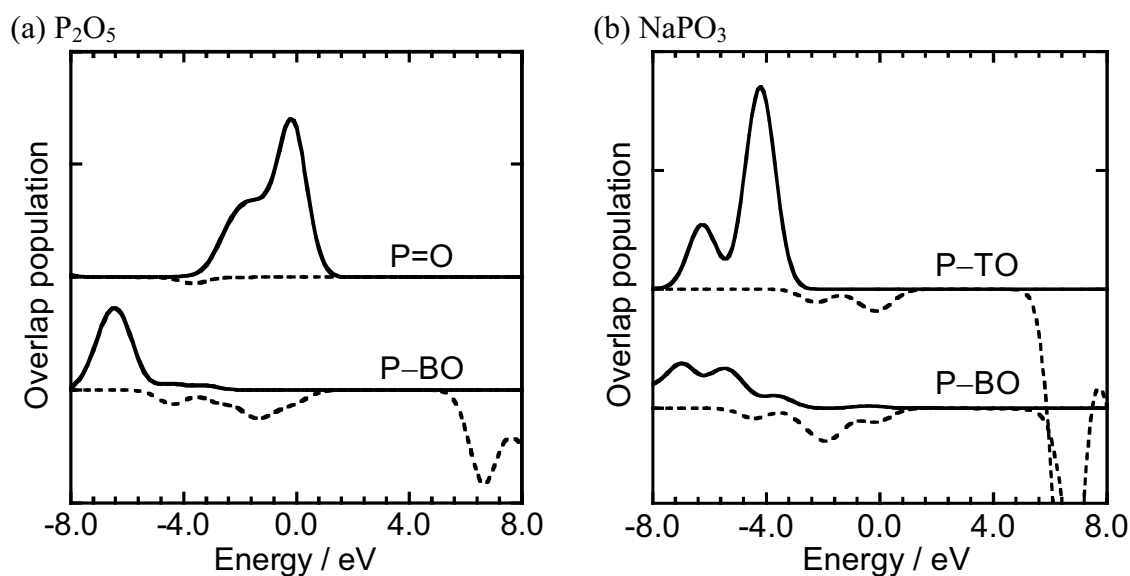


Figure 3: Overlap population diagrams for P–O bonds in some phosphate crystals.

(a)  $P_2O_5$ , (b)  $NaPO_3$  and (c)  $Na_5P_3O_{10}$ .

The continuous and broken lines indicate bonding and anti-binding overlap, respectively.

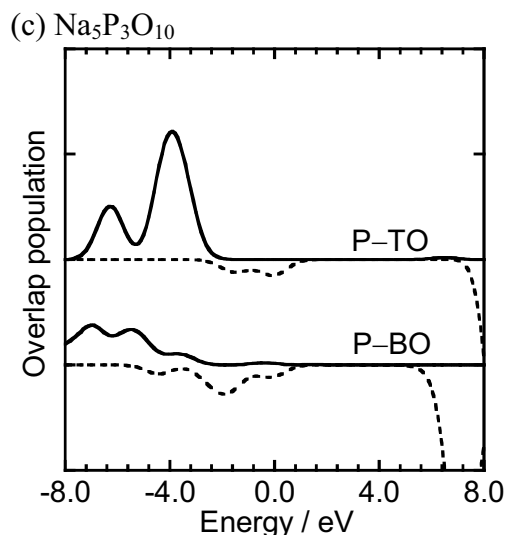


Figure 3: (continuation)

### Chemical Bonding Character of Aluminophosphates

The results of aluminophosphates are shown in Figure 4. It should be noted that the  $Q_{\text{MO}}$  values shown in Fig. 4 are obtained only from P–O and Al–O bonds in P–O–Al bridges, and  $Q_{\text{PO}}$  of P–O bonds in P–O–P bridges are not indicated in Fig. 4. With increasing Al/P atomic ratio,  $Q_{\text{MO}}$  of Al–O bonds increases and that of P–O bonds decreases. In general, tetrahedral and octahedral Al atoms have been regarded as network formers and modifiers, respectively, and hence oxygen atoms in P–O–Al(4) and P–O–Al(6) bridges are regarded as BOs and NBOs, respectively. Therefore, the result observed in Fig. 4, that is,  $Q_{\text{PO}}$  for P–O–Al(4) bridges in  $\text{AlPO}_4$  is smaller than that for P–O–Al(6) bridges in  $\text{NaAlP}_2\text{O}_7$  and  $\text{Al}(\text{PO}_3)_3$ , seems plausible.

In an aluminosilicate  $\text{KNa}_3\text{Al}_4\text{Si}_4\text{O}_{16}$  [18],  $Q_{\text{SiO}}$  and  $Q_{\text{AlO}}$  for Si–O–Al(4) bridges are 0.62 and 0.39, respectively. Comparing the P–O–Al(4) and Si–O–Al(4) bridges,  $Q_{\text{AlO}}$  are almost the same value, but  $Q_{\text{PO}}$  is much larger than  $Q_{\text{SiO}}$ . In  $\text{PO}_4$  units in  $\text{AlPO}_4$ , all oxygen atoms are in P–O–Al(4) bridges, and no terminal oxygen atoms are present. If  $\pi$ -bonding electrons originally localized on a P=O bond in a  $\text{PO}_4$  unit are evenly delocalized among the four P–O bonds, the bond order of P–O bonds will be 1.25. The larger bond order of P–O bonds leads to the larger  $Q_{\text{PO}}$  in  $\text{AlPO}_4$ .

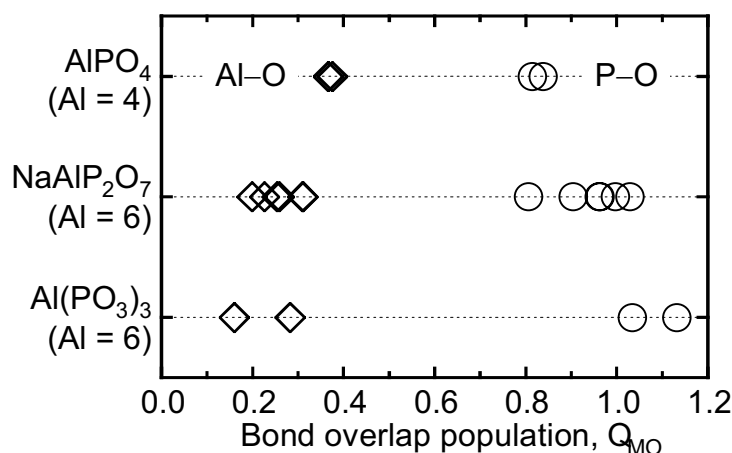


Figure 4: Bond overlap population,  $Q_{\text{MO}}$  of P–O and Al–O bonds in P–O–Al bridges in the aluminophosphate crystals.

### Chemical Bonding Character of Phosphates Containing Other Network Formers

It is well known that even the conventional network formers of B, Si and Ge atoms occupy higher coordination sites when small amount of the respective oxides are added to a phosphate glass. To the best of the authors' knowledge, B, Si and Ge atoms have higher coordination numbers in crystalline phosphates. Then, chemical bonding character of P–O–M bridges (M = B(4), Si(6) and Ge(6) ) was investigated. As shown in Figure 5,  $Q_{PO}$  is larger than  $Q_{MO}$  in every P–O–M bridge, which is probably due to the larger electronegativity of P (2.19) compared with the other network forming atoms (B = 2.04, Al = 1.61, Si = 1.90, Ge = 2.01), where Pauling electronegativities [20] are indicated. Electrons on P–O–M bridges gravitate toward more electronegative P atoms, resulting in larger  $Q_{PO}$  than  $Q_{MO}$ . Thus, less electrons are localized on M–O bonds in P–O–M bridges, and hence more M–O bonds are required to compensate the positive charge of central M atoms, resulting in the higher coordination numbers of M. As also shown in Fig. 5,  $Q_{PO}$  and  $Q_{MO}$  are ranked in ascending and descending orders of P–O–B(4), P–O–Si(6) and P–O–Ge(6), respectively, and the difference between  $Q_{PO}$  and  $Q_{MO}$  increases in that order. If the electronegativity of Ge were between Al and Si, the difference in bond overlap population would be in consistent with the difference in electronegativity. [20].

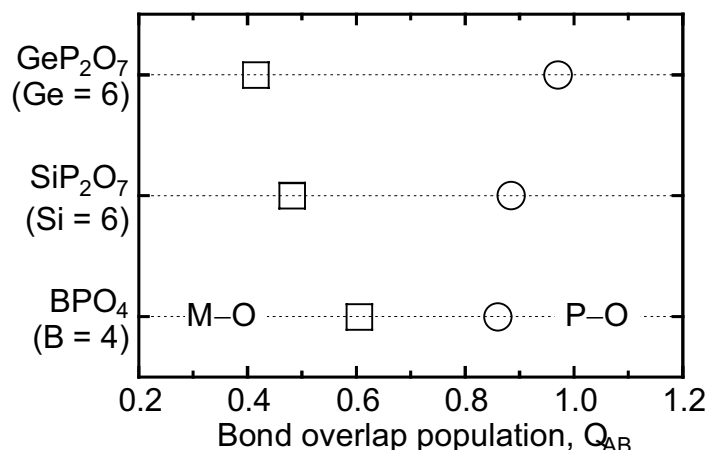


Figure 5: Bond overlap population,  $Q_{AB}$  of P–O and M–O bonds in P–O–M bridges in the phosphate crystals.

### **Conclusion**

Molecular orbital calculations have been performed to investigate the chemical bonding character of phosphate glasses, where cluster models were constructed based on the phosphate crystals. The chemical bonding character was evaluated by the bond overlap population  $Q_{AB}$ .  $Q_{PO}$  of P=O double bonds was much larger than those of terminal B–NBO and Si–NBO bonds, which was due to the larger bond order of P=O double bonds. By addition of alkali oxides,  $Q_{PO}$  of the terminal P–TO (P=O + P–NBO) bonds decreased, which was characteristic in phosphate system because in silicate system Si–NBO bonds increased in  $Q_{SiO}$  with increasing alkali content. According to overlap population diagram, anti-bonding overlap of Si and O orbitals in Si–NBO bonds was not observed at around HOMO levels, and in P–TO bonds, however, anti-bonding overlap was clearly seen, which probably lead to the unique change in  $Q_{PO}$  in the phosphate system. P–O–P bridging bonds in pure P<sub>2</sub>O<sub>5</sub> crystal indicated much highest  $Q_{AB}$  as compared with the bridging bonds in B<sub>2</sub>O<sub>3</sub>, SiO<sub>2</sub> and GeO<sub>2</sub> crystals, suggesting the highest bond covalency among the conventional glass formers.

As for P–O–Al bridges in aluminophosphates,  $Q_{PO}$  in P–O–Al(4) was smaller than that in P–O–Al(6),

suggesting the delocalization of electrons through tetrahedral Al atoms.  $Q_{PO}$  was higher than  $Q_{AlO}$  regardless of the coordination number of Al atoms. Such the disproportionate sharing of electrons in P–O–M bridges was also found in the cases of M = B(4), Si(6) and Ge(6) as well as Al(4) and Al(6). In every combination,  $Q_{PO}$  was always larger than  $Q_{MO}$ , which was responsible for larger electronegativity of P atom. Less  $Q_{MO}$  in P–O–M bridges was also responsible for the preferential occupation in higher coordination sites of M atoms in phosphate glasses.

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